

PAPER

van der Waals coefficients of the multi-layered MoS₂ with alkali metals




To cite this article: Shankar Dutt *et al* 2020 *Phys. Scr.* **95** 095506

View the [article online](#) for updates and enhancements.

You may also like

- [Ca₂C MXene monolayer as a superior anode for metal-ion batteries](#)
Kaptan Rajput, Vipin Kumar, Siby Thomas et al.
- [Long-range interactions between alkali and alkaline-earth atoms](#)
Jun Jiang, Yongjun Cheng and J Mitroy
- [Positron total scattering cross-sections for alkali atoms](#)
Nidhi Sinha, Suvam Singh and Bobby Antony

van der Waals coefficients of the multi-layered MoS₂ with alkali metals

Shankar Dutt¹ , Sukhjit Singh¹, A Mahajan¹, Bindiya Arora¹  and B K Sahoo² 

¹Department of Physics, Guru Nanak Dev University, Amritsar, Punjab-143005, India

²Atomic, Molecular and Optical Physics Division, Physical Research Laboratory, Navrangpura, Ahmedabad-380009, India

E-mail: bindiya.phy@ndu.ac.in

Received 19 May 2020, revised 6 August 2020

Accepted for publication 11 August 2020

Published 30 August 2020



CrossMark

Abstract

The van der Waals coefficients and the separation dependent retardation functions of the interactions between the atomically thin films of the multi-layered transition metal molybdenum disulfide (MoS₂) dichalcogenides with the alkali atoms are investigated. First, we determine the frequency-dependent dielectric permittivity and intrinsic carrier density values for different layers of MoS₂ by adopting various fitting models to the recently measured optical data reported by Yu and co-workers (2015 *Sci. Rep.* **5**, 16 996) using spectroscopy ellipsometry. Then, dynamic electric dipole polarizabilities of the alkali atoms are evaluated very accurately by employing the relativistic coupled-cluster theory. We also demonstrate the explicit change in the above coefficients for different number of layers. These studies are highly useful for the optoelectronics, sensing and storage applications using layered MoS₂.

Supplementary material for this article is available [online](#)

Keywords: van der Waals interaction, molybdenum disulfide dichalcogenide, dipole polarizability

(Some figures may appear in colour only in the online journal)

1. Introduction

Recent advances in the fabrication and synthesis of ultra-thin layered materials with unit cell thickness has boosted the art of continuously tailoring the properties of materials. Among these, graphene, a two dimensional (2D) material of carbon atoms, exhibit unique electronic, physical and chemical properties. However, zero band gap of graphene restricts the direct application of graphene in the electronic devices. This has prompted to search for composite graphene-like materials with a finite band gap. Transition metal dichalcogenides (TMDs) possessing the identical lamellar structure of graphite manifest remarkable applications in nano-electronics, sensors, catalytic, energy conversion and storage devices. The nature of transition elements present in these materials affects their structures. TMDs containing transition elements from the IV-VII groups of the periodic table exhibit layered structures,

while those containing transition elements belonging to the VIII-X groups show non-layered structures [1].

In addition to the homo-layer configurations in the 2D TMDs, the nanoscale heterostructures of TMDs have also been found to be suitable for the implementation of novel photonic and electronic devices [2, 3]. Theoretical studies demonstrate that several 2D TMDs offer a plethora of opportunities using lateral and vertical heterostructures due to their tunable broad-range optical bandgap and strong light-matter interactions [4, 5]. These heterostructures can be classified into three types on the basis of their band alignments; i.e. symmetric (type I), staggered (type II) and broken (type III) [6]. All these materials find many applications in high performance devices such as light-emitting diodes, photodetectors and transistors [7–14]. Moreover, strong Coulomb interactions and anisotropic dielectric environment lead to the formation of strongly bound excitons, trions, and biexcitons in these materials [15–17]. The hetero-bilayers of

TMDs with inter-layer excitons are emerging as novel class of long-lived dipolar composite bosons for optoelectronic applications [18]. Recently, it has been found that magnetic fields can promote the formation of biexciton to create favorable conditions for the formation of multiple exciton complexes, exciton super-fluidity, and biexciton condensates to materialize their practical applications [19].

Among TMDs, molybdenum disulfide (MoS_2) has emerged as one of the promising next-generation 2D materials with exceptional photonic, non-linear and electronic properties, in contrast to its bulk counterpart [20–22] and attracted applications among flexible gas sensing [23, 24] and optoelectronic devices [25–35]. In order to design and simulate the next-generation nanoelectronic devices built with MoS_2 , it is important to gain accurate knowledge of its electrical permittivity (ϵ), which is a fundamental property that characterizes refractive index, absorption, conductivity, capacitance, and many other intrinsic phenomena of a material [36]. Owing to complexity involved in the determination of ϵ values, a number of studies on these quantities for different layers of MoS_2 have been carried out. These investigations report a wide range of values with substantial differences in magnitudes from each other. This can be evident from the following: Liang *et al* [37] and Beal *et al* [38] presented a reflectivity spectrum of MoS_2 and calculated ϵ values by adopting Kramers-Kronig procedures. Liu *et al* [39] demonstrated that ϵ can be deduced from the absorption spectra. They first extracted out the imaginary part of ϵ from the absorption spectra, then estimated the real part using the Kramers-Kronig relation. Li *et al* [40] inferred ϵ from differential reflection spectra using an effective reflection coefficient method. Castellanos-Gomez *et al* [41] studied the refractive index of thin MoS_2 crystal with the Fresnel law and further predicted ϵ values. Recently, Yu *et al* [42] have measured ϵ values as functions of the number of layers for a discrete wavelength spectra in the visible region (345nm to 1000 nm) using spectroscopic ellipsometry technique. Their employed method is specially designed to measure optical data very accurately, so it is expected that measured values of ϵ by Yu *et al* are more reliable than the above-estimated values using various methods.

Apart from the electronic properties, knowledge of single atom adsorption with the atomically thin layered surfaces is of great importance for many practical applications. For instance, the alkali metal atom adsorption on graphene generally leads to an increase in its Fermi level, that has excellent potential for the field emission applications [43]. Moreover, Li ion storage capacity of single boron-doped graphene is found to be dramatically improved [44]. It is also known that if alkali atoms are absorbed on a metal surface, the electron and ion emission properties of the surface are drastically altered to provide improved applications in thermionics and physical electronics. The intercalations of alkali-metal ions (such as Li^+ , Na^+ , K^+) in 2D-layered MoS_2 can induce structural phase changes along with introducing changes in their electronic and optical properties [45–48]. The 2D MoS_2 nanoflakes on intercalation with Li^+ ions exhibit plasmon resonances near-UV and visible regions. These materials have

potential applications in the optoelectronics as well as in the plasmonic biosensing [49, 50]. Additional efforts have also been made to manipulate the electronic properties of MoS_2 through single-atom adsorption [51–53]. The van der Waals (vdW) interactions between atoms and material surfaces are critical for the study of physical adsorption. The interactions of atoms having lower ionization potentials with MoS_2 layers are considered to be crucial for a large number of possible applications requiring low-energy plasmas and ion beams [54]. From this point of view, it is important to fathom vdW interactions among alkali atoms with the material media; especially with the MoS_2 layers.

Motivated by the above developments, we report the vdW interactions between different alkali atoms and MoS_2 based TMDs. The electrical permittivity data required for such calculations have been taken from the ellipsometry measurements of Yu *et al* [42]. Calculations of dynamic dipole polarizabilities at imaginary frequencies for the respective atoms required for this study are carried out using relativistic coupled-cluster (RCC) theory. We qualitatively evaluate the intrinsic carrier density (N) for MoS_2 layers by fitting the experimental permittivity results with Drude-Lorentz (DL) oscillator. The DL model permits extraction of N from MoS_2 based TMDs for different number of layers. This allows us to examine the effect of N on the interaction coefficients as functions of the number of layers in MoS_2 . We find that the interactions between neutral atoms and MoS_2 are directly proportional to N , and they are maximum for monolayer. They decrease up to the 6th layer, thereafter they start increasing in the MoS_2 based TMDs.

The paper is organized as follows. In section 2, we present theoretical formulae used to calculate the vdW coefficients and the retardation functions, which can be used to describe the nature of the vdW interactions over a wide range of radial distance. In section 3, we discuss methods used for accurate evaluation of the atomic dynamic dipole polarizabilities. A well suited permittivity model to extract out the values of N is discussed in section 4. It follows by presenting results and discussion in section 5, before concluding in the last section.

2. Theory

A consistent theory, accounting for the electrical, mechanical and optical properties of materials, to study the vdW interactions among various atomic systems and real bodies made of different materials has been given by E. M. Lifshitz and collaborators [55, 56]. The atom-wall interactions can be computed by considering a polarizable particle interacting with a surface or a wall as a continuous medium having a frequency-dependent permittivity with real ($\epsilon_r(\omega)$) and imaginary ($\epsilon_i(\omega)$) parts. In this theory, the interaction potential of vdW interactions between an atom and a layered structure or a material plate can be efficiently described by the following

formula [55–58]

$$U(z) = -\frac{\alpha_{fs}^3}{2\pi} \int_0^\infty d\omega \omega^3 \alpha_n(\omega) \times \int_1^\infty d\xi e^{-2\alpha_{fs}\xi\omega z} H(\xi, \epsilon_r(\omega)), \quad (1)$$

where α_{fs} is the fine structure constant, z is the distance between the atom and the wall, and $\alpha_n(\omega)$ is the ground-state dynamic dipole polarizability of the atom with imaginary argument. The quantity $H(\xi, \epsilon_r(\omega))$, a function of Matsubara frequencies ξ and dielectric permittivity $\epsilon_r(\omega)$ of the material wall, is given by

$$H(\xi, \epsilon_r) = \left[(1 - 2\xi^2) \frac{\xi' - \epsilon_r \xi}{\xi' + \epsilon_r \xi} \right] + \frac{\xi' - \xi}{\xi' + \xi} \quad (2)$$

with $\xi' = \sqrt{\xi^2 + \epsilon_r - 1}$. The procedure for the evaluation of $\epsilon_r(\omega)$ is explained in [57–59]. In our study, the real ($n(\omega)$) and the imaginary ($\kappa(\omega)$) parts of the refractive index of MoS₂ are used to evaluate the imaginary parts of the dielectric permittivity of MoS₂ by the relation

$$\epsilon_i(\omega) = 2 n(\omega) \kappa(\omega). \quad (3)$$

We use the experimental values of $n(\omega)$ and $\kappa(\omega)$ from [42] to obtain the imaginary part of dielectric permittivity values. For conveniently carrying out the calculations and to predict the number of intrinsic carrier density N (electrons per unit volume) in the MoS₂ layers, we determine $\epsilon_i(\omega)$ using the DL oscillator model. This procedure has been discussed latter in detail. Further, we evaluate the real values of the dielectric permittivity at the imaginary frequencies by using the Kramers-Kronig formula [60]

$$\epsilon_r(\omega) = 1 + \frac{2}{\pi} \int_0^\infty d\omega' \frac{\omega' \epsilon_i(\omega')}{\omega^2 + \omega'^2}. \quad (4)$$

These values are calculated for the MoS₂ layers with layer number ranging from 1 to 10.

The vdW interaction potential can be conveniently expressed by [61]

$$U(z) = -\frac{C_3}{z^3} f_3(z), \quad (5)$$

where $f_3(z)$ is the retardation function and C_3 is known as the vdW coefficient, which is defined by

$$C_3 = \frac{1}{4\pi} \int_0^\infty d\omega \alpha_n(\omega) \vartheta(\omega) \quad (6)$$

with the factor

$$\vartheta(\omega) = \frac{\epsilon_r(\omega) - 1}{\epsilon_r(\omega) + 1}. \quad (7)$$

For the perfect conductor $\vartheta \rightarrow 1$, whereas for other materials, ϑ can be evaluated with the knowledge of their dielectric permittivities. By adopting a similar approach as in [58], we determine the vdW interaction potential between an atom and a thin layer of MoS₂ by using equation (1), and evaluate the C_3 coefficient using equation (6). By combining the C_3 coefficient and the interaction potential, the functional form of $f_3(z)$ for the vdW interaction potential is inferred from equation (5).

3. Dynamic polarizabilities of atoms

Evaluation of interaction potential $U(z)$ from equation (1) requires values of $\alpha_n(\omega)$. The procedure for determining accurate values of the dynamic polarizability of an atomic system having a closed core and a valence electron has been already described by us in [62, 63]. We apply the same procedure here to calculate the dynamic polarizabilities of the ground state of various alkali atoms considered in this study. In this approach, we divide the total dipole dynamic polarizability in terms of scalar and tensor components as follows

$$\alpha_n(\omega) = \alpha_n^{(0)}(\omega) + \frac{3M_{J_n}^2 - J_n(J_n + 1)}{J_n(2J_n - 1)} \alpha_n^{(2)}(\omega). \quad (8)$$

Here $\alpha_n^{(0)}(\omega)$ and $\alpha_n^{(2)}(\omega)$ are known as the scalar and tensor polarizabilities respectively. They are evaluated using the sum-over-states approach as

$$\alpha_n^{(0)}(\omega) = \sum_{k \neq n} W_n^{(0)} \left[\frac{|\langle \gamma_n J_n || \mathbf{D} || \gamma_k J_k \rangle|^2}{E_n - E_k + \omega} + \frac{|\langle \gamma_n J_n || \mathbf{D} || \gamma_k J_k \rangle|^2}{E_n - E_k - \omega} \right], \quad (9)$$

and

$$\alpha_n^{(2)}(\omega) = \sum_{k \neq n} W_{n,k}^{(2)} \left[\frac{|\langle \gamma_n J_n || \mathbf{D} || \gamma_k J_k \rangle|^2}{E_n - E_k + \omega} + \frac{|\langle \gamma_n J_n || \mathbf{D} || \gamma_k J_k \rangle|^2}{E_n - E_k - \omega} \right] \quad (10)$$

with the coefficients

$$W_n^{(0)} = -\frac{1}{3(2J_n + 1)}, \quad (11)$$

and

$$W_{n,k}^{(2)} = 2 \sqrt{\frac{5J_n(2J_n - 1)}{6(J_n + 1)(2J_n + 3)(2J_n + 1)}} \times (-1)^{J_n + J_k + 1} \begin{Bmatrix} J_n & 2 & J_n \\ 1 & J_k & 1 \end{Bmatrix}, \quad (12)$$

for the electric dipole (E1) reduced matrix elements $\langle \gamma_n J_n || \mathbf{D} || \gamma_k J_k \rangle$, J denotes the total angular momentum, E stands for energy and γ represents for the additional quantum numbers of atomic states.

For each component $i = 0$ and 2, we divide contributions to polarizability $\alpha_n^{(i)}$ into three parts, based on the correlation contributions from different types of electrons, as [62, 63]

$$\alpha_n^{(i)} = \alpha_{n,c}^{(i)} + \alpha_{n,cv}^{(i)} + \alpha_{n,v}^{(i)} \quad (13)$$

where $\alpha_{n,c}^{(i)}$, $\alpha_{n,cv}^{(i)}$ and $\alpha_{n,v}^{(i)}$ are referred to as the core, core-valence and valence correlation contributions, respectively. The $\alpha_{n,c}^{(i)}$ and $\alpha_{n,cv}^{(i)}$ contributions arise from the core-orbitals without considering and including interaction with valence orbital, respectively. These contributions are small in the alkali atoms. We, again, divide the $\alpha_{n,v}^{(i)}$ contribution into two parts; Main—containing dominant contributions from the low-lying excited states, and Tail—containing contributions

from the remaining excited states. As seen in the previous studies, major contributions to the polarizabilities of the atomic states in the alkali atoms come from $\alpha_{n,v}^{(i)}$ [59, 63–68] owing to the dominant contributions from the low-lying excited states. Evaluating the Main contribution exclusively has the advantage that uncertainty in its determination can be reduced by using excitation energies and reduced E1 matrix elements from the precise measurements wherever available. Contributions from the Tail part are estimated approximately using the Dirac-Fock (DF) method. Similarly, the core-valence contribution $\alpha_{n,cv}^{(0)}$ is also obtained using the DF method, whereas we adopt a relativistic random phase approximation, as discussed in [62, 69], to evaluate the $\alpha_{n,c}^{(0)}$ contribution.

For accurate evaluation of the E1 matrix elements involving the ground and low-lying excited states of the considered atoms, we employ the RCC theory ansatz. In this theory, the wave functions of atomic states in an alkali atom can be expressed by [70–75]

$$|\Psi_n\rangle = e^T \{1 + S_n\} |\Phi_n\rangle,$$

where $|\Phi_n\rangle = a_n^\dagger |\Phi_0\rangle$ with the DF wave function $|\Phi_0\rangle$ of the closed-core of the atom and a_n^\dagger denotes the valence orbital in a given state, T is known as the hole-particle excitation operator, which is responsible for exciting electrons from the occupied orbitals, and S_n corresponds to the excitation operator involving electron from the valence orbital n . In the present work, we have considered singles and doubles excitations in the RCC theory (RCCSD method) by expressing

$$T = T_1 + T_2 = \sum_{ap} a_p^\dagger a_a t_a^p + \frac{1}{4} \sum_{abpq} a_p^\dagger a_q^\dagger a_b a_a t_{ab}^{pq} \quad (14)$$

and

$$S_n = S_{1n} + S_{2n} = \sum_{n \neq p} a_p^\dagger a_n s_n^p + \frac{1}{2} \sum_{bpq} a_p^\dagger a_q^\dagger a_b a_n s_{nb}^{pq}, \quad (15)$$

where t_a^p and t_{ab}^{pq} are the amplitudes of the singles and doubles excitations of the T operator, respectively, and s_n^p and s_{nb}^{pq} are the amplitudes of the singles and doubles excitations of the S_n operator, respectively. After obtaining atomic wave functions in the RCCSD method, we calculate the E1 matrix element of a transition between the states $|\Psi_n\rangle$ and $|\Psi_k\rangle$ using the expression

$$\begin{aligned} \langle D \rangle_{nk} &\equiv \frac{\langle \Psi_n | D | \Psi_k \rangle}{\sqrt{\langle \Psi_n | \Psi_n \rangle \langle \Psi_k | \Psi_k \rangle}} \\ &= \frac{\langle \Phi_n | \tilde{D}_{nk} | \Phi_k \rangle}{\sqrt{\langle \Phi_n | \{1 + \tilde{N}_n\} | \Phi_n \rangle \langle \Phi_k | \{1 + \tilde{N}_k\} | \Phi_k \rangle}}, \end{aligned} \quad (16)$$

where $\tilde{D}_{nk} = \{1 + S_n^\dagger\} e^{T^\dagger} D e^T \{1 + S_k\}$ and $\tilde{N}_{i=n,k} = \{1 + S_i^\dagger\} e^{T^\dagger} e^T \{1 + S_i\}$. Calculation procedures of these expressions can be found elsewhere [70–75].

Table 1. Comparison of static polarizabilities (in a.u.) of the ground states of the Li, Na, K, Rb and Cs alkali atoms with their experimental values. Breakdown of different electron correlation effects for the determination of polarizabilities are also given explicitly.

	Li	Na	K	Rb	Cs
Main	162.5	161.4	284.3	309.4	382.9
Core	0.2	0.9	5.5	9.1	15.8
Valence-core	~0.0	~0.0	-0.1	-0.3	-0.5
Tail	~0.0	0.08	0.06	0.11	0.15
Total	162.7	162.3	289.8	318.5	398.4
Experimental	164.2 [80]	162.7 [81]	289.7 [82]	319.8 [82]	400.8 [82]

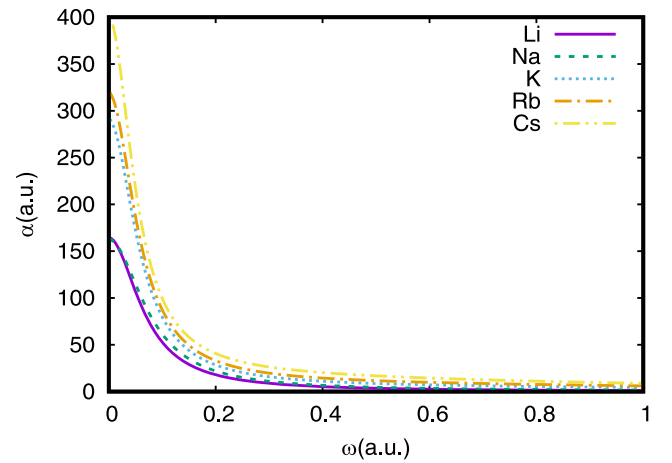


Figure 1. Plots showing dynamic polarizabilities (in a.u.) of the alkali Li, Na, K, Rb and Cs atoms in their ground state as function of frequency (in a.u.).

4. Models for permittivity determination

It is always desirable to have a logistic fit of the dielectric permittivity of a material media. For this purpose, a number of fitting models have been proposed in the literature [76, 77]. Drude developed a kinetic theory to account for the dielectric permittivity as well as its variation with frequency. In the Drude theory, the motion of a free electron in a material media can be described as a harmonic motion, where the electron oscillates under the influence of an electromagnetic wave. The oscillation leads to charge redistribution and create an additional induced electric field that restores electrons to their equilibrium positions. This back and forth periodic motion of electrons can be described mathematically by oscillators. Within this harmonic oscillator model, the frequency-dependent permittivity [78] can be presented as

$$\epsilon^D(\omega) = -\frac{\omega_p^2}{\omega^2 + i\gamma_d\omega}, \quad (17)$$

where ω_p is the plasma frequency relevant to the intraband transitions and can be written in terms of intrinsic carrier density N , reduced mass m^* and permittivity of free space ϵ_0

Table 2. Fitting parameters for the dynamic polarizabilities ($\alpha_n(\omega)$) of the considered alkali atoms. For unit conversion, one can use 1 a.u. of frequency $\omega = 27.21$ eV and 1 a.u. of $\alpha_n(\omega) = 0.2488319\text{kHz} (\text{kV cm}^{-1})^{-2}$.

Parameter	Atom									
	Li		Na		K		Rb		Cs	
	Frequency (ω) in a.u.									
	0 – 1.85	1.86 – 3000	0 – 2.24	2.25 – 3000	0 – 1.4	1.5 – 3000	0 – 1.05	1.06 – 3000	0 – 0.99	1 – 3000
α_0	0.442 77	0.0005	0.74571	0.00005	4.287 95	0.0022	7.26021	0.008 13	12.54507	0.013 91
ω_c	-0.00032	-0.91386	-0.00013	-1.82586	-0.00028	-0.64706	-0.00041	-0.70395	-0.00062	-0.6725
w	0.13766	1.759 99	0.1555	2.837 11	0.1201	1.50212	0.118 15	1.123 51	0.108 12	1.2767
A	35.392 79	11.556 82	39.472 51	28.468 48	53.872 49	60.89532	57.797 95	110.355 17	65.745 28	137.662 87

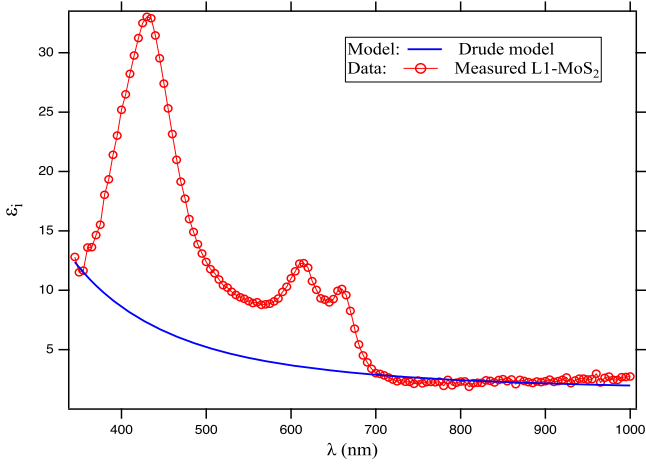


Figure 2. A comparative analysis of the imaginary part of the permittivity of the monolayer MoS₂ film estimated using the Drude model as given by equation (23) (blue curve) and the measured spectra from [42] (red circles).

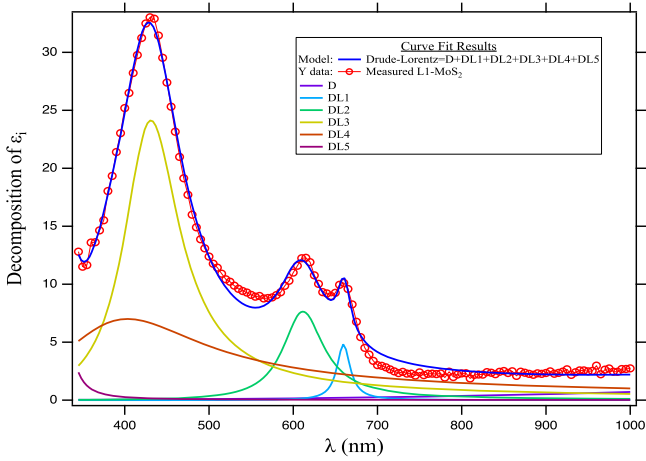


Figure 3. Plots of the imaginary parts of the dynamic permittivity values of the monolayer MoS₂ film estimated using the DL model given by equation (24) (blue curve) and the measured spectra from [42] (red circles) against wavelength (in nm). The ϵ_i values are decomposed into six components. The first component is named as ‘D’ corresponding to the first term of equation (24), whereas the other five components marked as ‘DL’ corresponding to $j = 1, 2, 3, 4$ and 5 in the summation of equation (24).

as

$$\omega_P = \frac{Ne^2}{\epsilon_0 m^*}. \quad (18)$$

Physically, the electromagnetic response of a material at ω_P changes from metallic to dielectric. γ_d in equation (17) is the damping coefficient, which describes the damping force arising due to subsequent collisions of electrons and is expressed as

$$\gamma_d = \frac{e}{m^* \mu}, \quad (19)$$

where μ is the carrier mobility and e is the electron charge. In our calculations, its value is taken to be $0.041 \text{ m}^2 \text{V}^{-1} \text{s}^{-1}$ [79]. The Drude model describes contributions only from the free

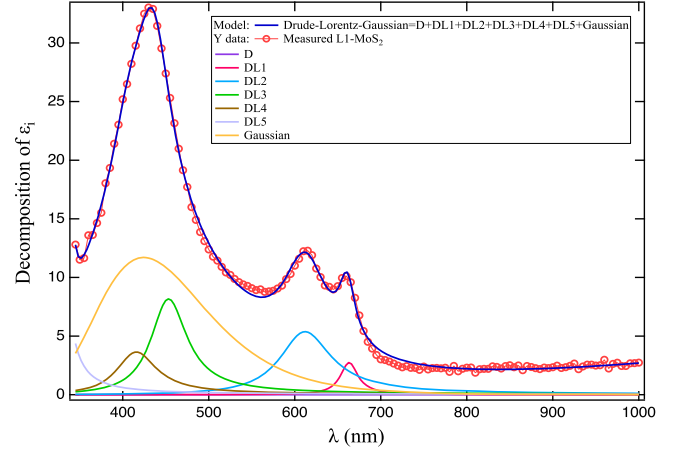


Figure 4. Plots of the imaginary parts of the permittivity of the monolayer MoS₂ film estimated by the LDG model given by equation (25) (blue curve) and from the measured spectra of [42] (red circles) against wavelength (in nm). Here, the ϵ_i is decomposed into seven components. The first component named as ‘D’ corresponding to $j = 0$ term of equation (25), whereas the next five components marked as ‘DL’ corresponding to $i = 1, 2, 3, 4$ and 5 . The last term corresponds to the Gaussian background added.

electrons to the permittivity, but it does not take into account the interband transitions of the bound electrons excited by the photons with higher energy. The contributions from these higher level interband electronic transitions to the dielectric permittivity can be expressed as a superposition of the Lorentz oscillators, given by

$$\epsilon^L(\omega) = \sum_{j=1}^5 \frac{f_j \omega_P^2}{\omega_j^2 - \omega^2 - i\gamma_j \omega}, \quad (20)$$

where j stands for the resonant nodes, ω_j corresponds to the resonance frequencies, f_j refers to the weighting factor and γ_j is the damping coefficient. It is worth noting that the Lorentz model reduces to the Drude Model for $j = 0$, $\omega_0 = 0$, $f_j = 1$ and $\gamma_0 = \gamma_d$.

In real materials, both free and bound electrons contribute to the dielectric permittivity. Therefore, the complete model contains both Drude component for intra band effect and Lorentz contribution for interband transitions. Accounting for them, the comprehensive DL model is represented as

$$\epsilon^{DL}(\omega) = \epsilon_\infty + \epsilon^D(\omega) + \epsilon^L(\omega), \quad (21)$$

where ϵ_∞ is the permittivity at $\omega \rightarrow \infty$, denoting the constant offset value. We have used this model to fit the available experimental values given in [42], then infer values at other frequencies for their applications.

5. Results and discussion

For realizing interactions between the multi-layered molybdenum disulfide with the alkali atoms, we require accurate values of dynamic polarizabilities of the alkali atoms. To validate the rigid correctness of these values, we first determine the static polarizabilities for the ground state of the considered alkali atoms and compare them with the available

Table 3. Weight factor f_j (dimensionless), damping coefficient γ_j (in eV), and ω_j resonance frequencies (in eV) for the Lorentz oscillators used in equation (24) for layers 1, 2 and 3. All the coefficients are normalized with \hbar .

j	Layer 1			Layer 2			Layer 3		
	$f_j/\hbar [\times 10^5]$	$\gamma_j/\hbar [\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar [\times 10^5]$	$\gamma_j/\hbar [\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar [\times 10^5]$	$\gamma_j/\hbar [\times 10^{-1}]$	ω_j/\hbar
1	0.25 ± 0.03	0.53 ± 0.04	1.877 ± 0.002	0.28 ± 0.01	0.64 ± 0.08	1.869 ± 0.002	0.21 ± 0.05	0.68 ± 0.03	1.867 ± 0.001
2	1.85 ± 0.22	2.22 ± 0.13	2.034 ± 0.002	1.51 ± 0.16	2.15 ± 0.12	2.027 ± 0.001	1.26 ± 0.07	2.14 ± 0.09	2.035 ± 0.002
3	28.81 ± 0.17	6.22 ± 0.09	2.895 ± 0.004	20.39 ± 0.31	5.61 ± 0.07	2.858 ± 0.003	15.55 ± 0.36	5.48 ± 0.14	2.812 ± 0.004
4	1.31 ± 0.24	2.89 ± 0.15	3.19 ± 0.03	5.48 ± 0.83	4.27 ± 0.24	3.16 ± 0.03	6.80 ± 0.91	5.04 ± 0.19	3.15 ± 0.04
5	1.23 ± 0.72	6.80 ± 0.94	3.80 ± 0.48	3.6 ± 1.2	3.30 ± 0.66	3.76 ± 0.41	3.46 ± 0.98	3.9 ± 1.3	3.67 ± 0.48

Table 4. Weight factor f_j (dimensionless), damping coefficient γ_j (in eV), and ω_j resonance frequencies (in eV) for the Lorentz oscillators used in equation (24) for layers 4, 5 and 6. All the coefficients are normalized with \hbar .

j	Layer 4			Layer 5			Layer 6		
	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar
1	0.20 ± 0.05	0.63 ± 0.06	1.866 ± 0.001	0.18 ± 0.06	0.52 ± 0.09	1.859 ± 0.001	0.19 ± 0.06	0.58 ± 0.10	1.871 ± 0.002
2	1.43 ± 0.13	2.16 ± 0.17	2.038 ± 0.001	1.37 ± 0.17	2.11 ± 0.21	2.028 ± 0.002	1.39 ± 0.16	2.08 ± 0.22	2.036 ± 0.003
3	16.85 ± 0.19	5.44 ± 0.11	2.794 ± 0.004	15.03 ± 0.42	5.29 ± 0.18	2.761 ± 0.002	15.20 ± 0.34	5.03 ± 0.14	2.757 ± 0.005
4	7.63 ± 0.82	4.95 ± 0.18	3.13 ± 0.06	8.12 ± 0.85	5.03 ± 0.22	3.10 ± 0.04	8.68 ± 0.77	4.78 ± 0.19	3.07 ± 0.04
5	5.6 ± 2.2	4.6 ± 1.4	3.62 ± 0.43	4.9 ± 1.9	4.6 ± 1.4	3.57 ± 0.51	5.0 ± 2.0	4.4 ± 1.4	3.56 ± 0.46

Table 5. Weight factor f_j (dimensionless), damping coefficient γ_j (in eV), and ω_j resonance frequencies (in eV) for the Lorentz oscillators used in equation (24) for layers 7, 8 and 9. All the coefficients are normalized with \hbar .

j	Layer 7			Layer 8			Layer 9		
	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar
1	0.23 ± 0.05	0.58 ± 0.05	1.871 ± 0.001	0.22 ± 0.04	0.59 ± 0.11	1.867 ± 0.002	0.28 ± 0.07	0.64 ± 0.13	1.873 ± 0.004
2	1.52 ± 0.11	2.05 ± 0.21	2.038 ± 0.003	1.62 ± 0.11	2.06 ± 0.26	2.042 ± 0.002	1.85 ± 0.13	2.13 ± 0.31	2.039 ± 0.004
3	17.05 ± 0.29	5.10 ± 0.21	2.749 ± 0.004	17.68 ± 0.29	5.07 ± 0.31	2.744 ± 0.004	19.98 ± 0.31	5.20 ± 0.39	2.721 ± 0.005
4	8.43 ± 0.93	4.73 ± 0.31	3.08 ± 0.05	8.80 ± 1.07	4.80 ± 0.25	3.06 ± 0.06	7.98 ± 0.97	4.79 ± 0.41	3.05 ± 0.05
5	5.0 ± 2.0	4.4 ± 1.5	3.55 ± 0.48	5.0 ± 1.8	4.4 ± 1.4	3.55 ± 0.51	5.1 ± 1.8	4.4 ± 1.4	3.56 ± 0.72

Table 6. Weight factor f_j (dimensionless), damping coefficient γ_j (in eV), and ω_j resonance frequencies (in eV) for the Lorentz oscillators used in equation (24) for layers 9 and 10. All the coefficients are normalized with \hbar .

j	Layer 10		
	$f_j/\hbar[\times 10^5]$	$\gamma_j/\hbar[\times 10^{-1}]$	ω_j/\hbar
1	0.25 ± 0.06	0.55 ± 0.08	1.861 ± 0.004
2	1.68 ± 0.12	2.01 ± 0.26	2.037 ± 0.005
3	18.17 ± 0.34	4.83 ± 0.34	2.72 ± 0.005
4	7.40 ± 1.10	4.59 ± 0.31	3.05 ± 0.05
5	3.4 ± 1.9	4.1 ± 1.6	3.56 ± 0.55

measurements. Our final calculated polarizability values along with the contributions from the core, core-valence and valence correlations are tabulated in table 1. As can be seen, our calculated value of the ground state of Li is 162.7 a.u., which is in good agreement with the polarizability value of 164.2 a.u. measured by Miffre *et al* [80] using atom interferometry. Similarly, our estimated value for Na atom is 162.3 a.u. against its experimental result 162.7 a.u. reported by Holmgren *et al* [81]. The values obtained for K, Rb and Cs atoms from our calculations are 289.8 a.u., 318.5 a.u. and 398.4 a.u., respectively. These values are also in good agreement with available measurements [82]. This demonstrates that the dynamic dipole polarizabilities of the investigated alkali atoms can be determined with sub-one percent accuracy for the intended study.

We plot the dynamic polarizabilities obtained by us for the alkali atoms in figure 1. To infer their values at a particular frequency, we provide a fitting formula as

$$\alpha(\omega) = \alpha_0 + \frac{2A}{\pi} \frac{w}{4(\omega - \omega_c)^2 + w^2}, \quad (22)$$

where α_0 , A , w and ω_c are the fitting parameters. These parameters depend on the atom and range of frequency. We provide these fitting parameters in table 2 for two different ranges of frequency to extrapolate the results.

We use the previously discussed models to fit the dynamic values of permittivity available in literature [42] and recommend the best fitted permittivity model for MoS₂ layers. We consider only the imaginary part of the permittivity as the real part can be estimated using the Kramer-Kronig relation, given by equation (4). The formulae for the imaginary part of ϵ following equations (17) and (20), are given by

$$\epsilon_i^D(\omega) = \frac{\gamma_d \omega_p^2}{\omega(\omega^2 + \gamma_d^2)} \quad (23)$$

and

$$\epsilon_i^{DL}(\omega) = \frac{\alpha \gamma_d \omega_p^2}{\omega(\omega^2 + \gamma_d^2)} + \sum_{j=1}^5 \frac{f_j \gamma_j \omega \omega_p^2}{\gamma_j^2 \omega^2 + (\omega^2 - \omega_j^2)^2} \quad (24)$$

in the Drude and DL models, respectively. The $\epsilon_i^D(\omega)$ values of monolayer MoS₂ using equation (23) have been graphically presented in figure 2. As can be verified from the figure that the Drude model gives accurate permittivity values only

in the infrared region and the experimental data disagrees with predictions from Drude model in the visible wavelength (shorter than 700 nm). In this region various interband transitions start contributing. Therefore, it is expected that the DL model will provide a better fit to the measured values. The $\epsilon_i^{DL}(\omega)$ values using equation (24) along with the experimental permittivity values are shown in figure 3. It can be noticed from this figure that the measured data from [42] is consistent with the results estimated by the DL model. This suggests that the values estimated using the DL model can be assumed to be reliable for further analysis. The authors in [77] have used a hybrid Lorentz-Drude-Gaussian (LDG) model in their study to fit the permittivity data for monolayer of MoS₂, which is given by

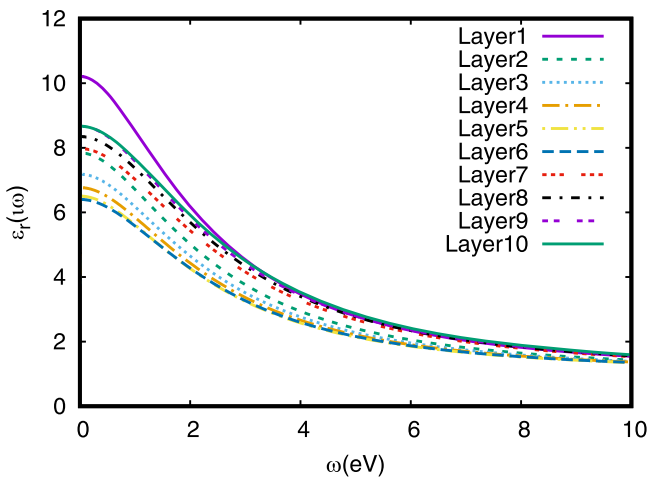
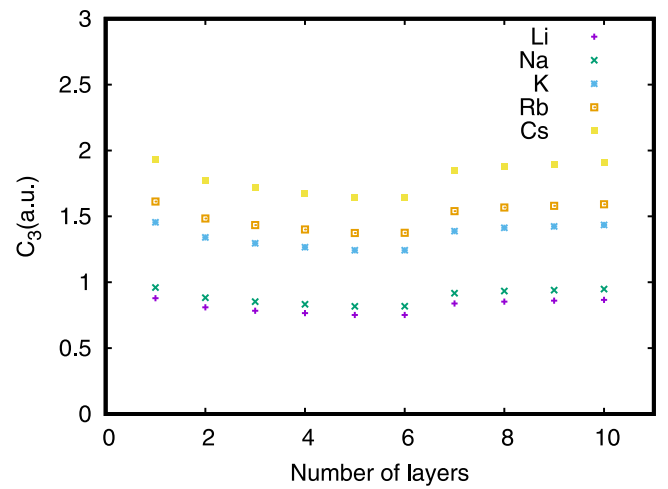
$$\epsilon_i^{LDG}(\omega) = \epsilon_\infty + \sum_{j=0}^5 \frac{f_j \gamma_j \omega_p^2 \omega}{(\omega_j^2 - \omega^2)^2 + \omega^2 \gamma_j^2} + \eta \exp\left(-\frac{(\hbar\omega - \beta)^2}{2\sigma^2}\right), \quad (25)$$

with β as mean, σ as variance and η as the maximum amplitude of the Gaussian function. In this case, the term with $j = 0$ and $\omega_0 = 0$ carries a weight factor $f_0 \neq 1$. As seen in figure 4, by adding a Gaussian background of the above kind with our DL model does not bring much change to our fitted values. Since there is no physical interpretation of the Gaussian component and it is added only as a background to match the estimated values with the experimental results, this justifies our above assertion that the DL model is able to predict permittivity values accurately.

In tables 3, 4, 5, 6, and 7 we present the fitting values of γ_d , ω_p , γ_j , ω_j and f_j from the DL model for various layers of MoS₂ along with the uncertainties in them. We use a non-linear least-squares minimization technique to extrapolate the permittivity values at different frequencies and they are decomposed into six components. The first component is named as 'D' corresponding to the first term of equation (24), whereas the other five components marked as 'DL_j' corresponding to $j = 1, 2, 3, 4$ and 5 in the summation of equation (24). We have provided a code written using the python programming language in the Supplementary Material available online at stacks.iop.org/PS/95/095506/mmedia that is used for carrying out this fitting. As can be seen from table 3, our ω_j/\hbar values for monolayer agree well with those predicted in literature [77, 83–87]. We also note from these tables that although the uncertainties are quite small for the DL1, DL2, DL3 and DL4 components, they are quite significant for the DL5 and D components. For the DL5 component, only a few data points are available for fitting which lead to significant uncertainties in the fitting parameters corresponding to this component. Also, the inferred ω_p value of 27.59 meV for monolayer of MoS₂, given in table 7, matches very well with the measured plasma frequency of 28.3 meV [88]. We use the fitting value of γ_d in equation (19) to estimate the effective mass m^* for various number of MoS₂ layers. From these calculated m^* values, we further evaluate intrinsic carrier density N using the fitting values of ω_p in equation (18). From table 7, we also note that N is maximum

Table 7. Values of the plasma frequency ω_p , damping coefficient γ_d and the calculated intrinsic carrier density N for layer numbers 1 to 10. Here m_0 is the mass of an electron.

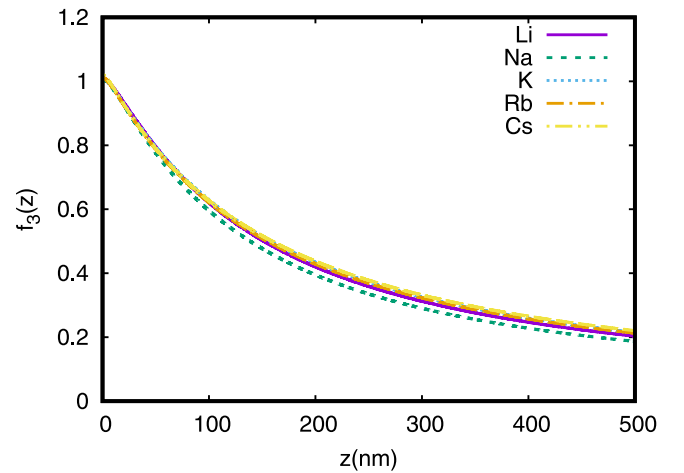
Layer number	ω_p (in meV)	γ_d (in $\times 10^{-2}$ eV)	m^* (in m_0)	N (in $\times 10^{15}$ cm $^{-3}$)
1	27.59 \pm 0.02	3.10 \pm 0.93	0.57 \pm 0.30	1.20 \pm 0.63
2	26.68 \pm 0.02	3.18 \pm 0.81	0.56 \pm 0.25	1.13 \pm 0.52
3	25.59 \pm 0.01	3.20 \pm 0.78	0.55 \pm 0.24	1.08 \pm 0.48
4	25.16 \pm 0.02	3.26 \pm 0.82	0.54 \pm 0.25	1.04 \pm 0.49
5	25.06 \pm 0.03	3.31 \pm 0.86	0.54 \pm 0.26	1.02 \pm 0.49
6	24.81 \pm 0.02	3.42 \pm 0.77	0.52 \pm 0.23	0.98 \pm 0.43
7	25.31 \pm 0.04	3.39 \pm 1.05	0.52 \pm 0.31	1.01 \pm 0.60
8	25.63 \pm 0.02	3.33 \pm 0.81	0.53 \pm 0.24	1.04 \pm 0.48
9	25.96 \pm 0.02	3.28 \pm 0.94	0.54 \pm 0.29	1.07 \pm 0.56
10	26.34 \pm 0.04	3.20 \pm 0.77	0.55 \pm 0.24	1.11 \pm 0.49

**Figure 5.** Plots of real parts of the dielectric permittivity at imaginary frequency as function of frequencies (in eV) for different number of MoS₂ layers.**Figure 6.** Plots showing the C_3 coefficients for Li, Na, K, Rb and Cs atoms with varying layer numbers of MoS₂.**Table 8.** Calculated C_3 coefficients (in a.u.) for interaction between different layers of MoS₂ with the alkali-metal atoms.

Layer	Li	Na	K	Rb	Cs
1	0.879	0.960	1.455	1.613	1.932
2	0.810	0.883	1.340	1.484	1.776
3	0.784	0.853	1.296	1.434	1.716
4	0.766	0.833	1.266	1.401	1.677
5	0.752	0.817	1.243	1.374	1.645
6	0.752	0.818	1.243	1.375	1.645
7	0.839	0.917	1.388	1.540	1.845
8	0.853	0.933	1.413	1.567	1.878
9	0.860	0.940	1.424	1.580	1.893
10	0.866	0.948	1.435	1.592	1.908

for a monolayer of MoS₂, thereafter, it starts decreasing up to layer number 6 and starts increasing again as the number of layers are increased up to 10.

Next, we find $\epsilon_r(\omega)$ values extracted by substituting $\epsilon_r(\omega)$ values in equation (4) for different layers of MoS₂, which are plotted against frequency in figure 5. The behaviour of $\epsilon_r(\omega)$ as a function of layer number is seen to be in accordance with

**Figure 7.** The retardation coefficients ($f_3(z)$) for Li, Na, K, Rb and Cs atom as functions of the atom-wall separation distance z .

the observation by Yu *et al* [42]. In their work, the authors demonstrate that excitonic effects play a dominant role in the dielectric function of 5-7 layered MoS₂. Therefore, the dielectric function decreases with the layer number up to 6 but turns to increase further with the increase in layer number.

The C_3 coefficients for the interactions between Li, Na, K, Rb and Cs atoms and the MoS₂ layers evaluated using $\epsilon_r(\omega)$ values are listed in table 8. A comparison of the C_3 coefficient as a function of layer number reveals that the interaction is maximum between atoms and monolayer of MoS₂. The interaction decreases with an increase in the number of MoS₂ layers up to the sixth layer, then it starts increasing again. The trend is found to be common for all the considered atoms. It is also quite evident that the trend followed by the C_3 coefficients with increasing number of layers is similar to that predicted for the intrinsic carrier density N . This observation is explained using the fact that the strength of the vdW force depends on the electric polarizability of the interacting atom. The tendency of the MoS₂ layer to polarize the incoming atom increases with the increase in the number of electrons per unit volume. As a result, the values of C_3 see an upsurge with an escalation in N .

A graphical representation for the C_3 coefficients for the considered alkali atoms with varying layer number is shown in figure 6. Our results in this figure support the finding that for the same layer number, the C_3 coefficients increase with increase in the atomic number. We also notice that the ratio of C_3 coefficients among various atoms vary slowly with the number of layers. For instance, the ratio of the C_3 coefficient for the interaction of any layer of MoS₂ with Rb and Li atoms is 1.83 irrespective of the number of layers. This knowledge of variation pattern of C_3 coefficients with number of layers with different alkali atoms will pave way to design sensors for detecting the alkali atoms by the MoS₂ layers. To give an estimate of these interactions at an intermediate distance, we next calculate the retardation function $f_3(z)$ as a function of distance z for various number of layers in MoS₂ based TMDs. We have shown comparison of the $f_3(z)$ values between an atom and the MoS₂ monolayer in figure 7. It is clear from this figure that the retardation function decreases with increase in z . Also, we notice from the above figure that the retardation function is similar for all the considered atoms, and it is not affected much with the atomic size.

6. Conclusion




To summarize, we have investigated the C_3 coefficients for the interactions between the alkali atoms with the MoS₂ layers. We performed high accuracy calculations of dynamic dipole polarizability of the considered alkali atoms and determined the dynamic dielectric permittivities for different layers of MoS₂ over a wide range of imaginary frequency. We have proposed a readily usable logistic fit for the dielectric permittivity for various layers of MoS₂ ranging from 1 to 10. We have also shown dependency of the intrinsic carrier density N and the coefficients with increasing layer numbers of the MoS₂ surface. Variation of C_3 as well as N with the number of layers shows decrease in values up to 6 number of layers. This finding could be useful for the formation of highly sensitive and reproducible sensing probes for detection of alkali atoms using 1-6 layered MoS₂ based transition metal dichalcogenides. Our study reveals that the ratios of the C_3

coefficients among various atoms do not change as the layer number is changed.

Acknowledgments

The work of B.A. is supported by the DST-SERB Grant No. EMR/2016/001228.

ORCID iDs

Shankar Dutt  <https://orcid.org/0000-0002-6814-070X>
 Bindiya Arora  <https://orcid.org/0000-0001-7083-034X>
 B K Sahoo  <https://orcid.org/0000-0003-4397-7965>

References

- [1] Han S A, Bhatia R and Kim S 2015 *Nano Convergence* **2** 17
- [2] Gong C et al 2013 *Appl. Phys. Lett.* **103** 053513
- [3] Komsa H and Krasheninnikov A 2013 *Phys. Rev. B* **88** 085318
- [4] Sahoo P K et al 2019 *ACS Nano* **13** 12372–84
- [5] Sahoo P K, Memaran S, Xin Y, Balicas L L and Gutiérrez H R 2018 *Nature* **553** 63–7
- [6] Özçelik V O, Azadani J G, Yang C, Koester S J and Low T 2016 *Phys. Rev. B* **94** 035125
- [7] Withers F et al 2015 *Nat. Mater.* **14** 301–6
- [8] Xu W et al 2017 *Nature* **541** 62–7
- [9] Nakamura S et al 1995 *J. Appl. Phys.* **34** L797
- [10] Yu W J et al 2013 *Nat. Nanotechnol.* **8** 952–8
- [11] Zhang Y et al 2011 *IEEE J. Quantum Electron.* **47** 1475–9
- [12] Sarkar D et al 2015 *Nature* **526** 91–5
- [13] Lin Y C et al 2015 *Nat. Commun.* **6** 7311
- [14] Koswatta S O, Koester S J and Haensch W 2010 *IEEE Transactions on Electronic Devices* **57** 3222–30
- [15] You Y et al 2015 *Nat. Phys.* **11** 477–81
- [16] He K et al 2014 *Phys. Rev. Lett.* **113** 026803
- [17] Mak K F et al 2013 *Nat. Mater.* **12** 207–11
- [18] Kremser M et al 2020 *npj 2D materials and Applications* **4** 8
- [19] Stevens C E et al 2018 *Nat. Commun.* **9** 3720
- [20] Kourosh K Z, Wan Q H, Kis A, Coleman J N and Strano M S 2012 *Nat. Nanotechnol.* **7** 699
- [21] Chhowalla M, Shin H S, Eda G, Li L J, Loh K P and Zhang H 2013 *Nat. Chem.* **5** 263
- [22] Li X and Zhu H 2015 *Journal of Materiomics* **1** 33–44
- [23] Donarelli M and Ottaviano L 2018 *Sensors (Basel, Switzerland)* **18** 3638
- [24] Akbari E, Jahanbin K, Afroozeh A, Yupapin P and Buntat Z 2018 *Physica B* **545** 510–8
- [25] Mak K F, Lee C, Hone J, Shan J and Heinz T F 2010 *Phys. Rev. Lett.* **105** 136805
- [26] Brasch V, Geiselmann M, Herr T, Lihachev G, Pfeiffer M H, Gorodetsky M L and Kippenberg T J 2016 *Science* **351** 357
- [27] Tsai M L, Su S H, Chang J K, Tsai D S, Chen C H, Wu C I, Li L J, Chen L J and He H Jr. 2014 *ACS Nano* **8** 8317–22
- [28] Yin Z, Li H, Li H, Jiang L, Shi Y, Sun Y, Lu G, Zhang Q, Chen X and Zhang H 2012 *ACS Nano* **6** 74–80
- [29] Guo X, Zou C-L, Jung H and Tang H X 2016 *Phys. Rev. Lett.* **117** 123902
- [30] Tongay S, Zhou J, Ataca C, Lo K, Matthews T S, Li J, Grossman J C and Wu J 2012 *Nano Lett.* **12** 5576
- [31] Gutierrez H R, Perea-Lopez N, Elias A L, Berkdemir A, Wang B, Lopez-Urias R L F, Crespi V H, Terrones H and Terrones M 2013 *Nano Lett.* **13** 3447

- [32] Sundaram R S, Engel M, Lombardo A, Krupke R, Ferrari A C, Avouris P and Steiner M 2013 *Nano Lett.* **13** 1416
- [33] Zhao W, Ghorannevis Z, Chu L, Toh M, Kloc C, Tan P H and Eda G 2013 *ACS Nano* **7** 791
- [34] Krishnan U, Kaur M, Singh K, Kumar M and Kumar A 2019 *Superlattices Microstruct.* **128** 274–97
- [35] Li Z, Meng X and Zhang Z 2018 *J. Photochem. Photobiol., C* **35** 39–55
- [36] Santos E J G and Kaxiras E 2013 *ACS Nano* **7** 10741–6 PMID: 24 215 099
- [37] Liang W Y 1971 *J. Phys. C: Solid State Phys.* **4** L378–81
- [38] Beal A R and Hughes H P 1979 *J. Phys. C: Solid State Phys.* **12** 881–90
- [39] Liu J-T, Wang T-B, Li X-J and Liu N-H 2014 *J. Appl. Phys.* **115** 193511
- [40] Li S L, Song H, Kuramochi H, Nakaharai S and Tsukagoshi K 2012 *ACS Nano* **8** 7381
- [41] Castellanos-Gomez A, Agraït N and Rubio-Bollinger G 2010 *Appl. Phys. Lett.* **96** 213116
- [42] Yu Y et al 2015 *Sci. Rep.* **5** 16996
- [43] Sahin H and Peeters F M 2013 *Phys. Rev. B* **87** 085423
- [44] Wang X, Zeng Z, Ahn H and Wang G 2009 *Appl. Phys. Lett.* **95** 183103
- [45] He Q Y et al 2019 *Nano Lett.* **19** 6819–26
- [46] Jung Y, Zhou Y and Cha J J 2016 *Inorg. Chem. Front.* **3** 452–63
- [47] Xiong F et al 2015 *Nano Lett.* **15** 6777–84
- [48] Xia J et al 2017 *Nanoscale* **9** 7533–40
- [49] Wang Y C et al 2013 *ACS Nano* **7** 10083–93
- [50] Wang Y C et al 2015 *Nano Lett.* **15** 883–90
- [51] Li X D, Fang Y M, Wu S Q and Zhu Z Z 2015 *AIP Adv.* **5** 057143
- [52] Xiao D, Liu G B, Feng W, Xu X and Yao W 2012 *Phys. Rev. Lett.* **108** 196802
- [53] Ataca C and Ciraci S 2011 *J. Phys. Chem. C* **115** 13303
- [54] Gadzuk J W 1967 Theory of atom-metal interactions *Surf. Sci.* **6** 133–58
- [55] Lifshitz E M and Pitaevskii L P 1980 *Statistical Physics* (Oxford: Pergamon)
- [56] Lifshitz E M 1955 *Zh. Exsp. Toer. Fiz.* **29** 94
- [57] Lach G, Dekieviet M and Jentschura U D 2010 *Int. J. Mod. Phys. A* **25** 2337
- [58] Arora B, Kaur H and Sahoo B K 2014 *J. Phys. B: At. Mol. Opt. Phys.* **47** 155002
- [59] Arora B and Sahoo B K 2012 *Phys. Rev. A* **86** 033416
- [60] Melrose D B and Stoneham R J 1977 *J. Phys. A: Math. Gen.* **10** L17-L20
- [61] Kharchenko P, Babb J F and Dalgarno A 1997 *Phys. Rev. A* **55** 3566
- [62] Kaur J, Nandy D K, Arora B and Sahoo B K 2015 *Phys. Rev. A* **91** 012705
- [63] Arora B, Nandy D K and Sahoo B K 2012 *Phys. Rev. A* **85** 012506
- [64] Derevianko A, Johnson W R, Safronova M S and Babb J F 1999 *Phys. Rev. Lett.* **82** 3589
- [65] Arora B, Safronova M S and Clark C W 2007 *Phys. Rev. A* **76** 052509
- [66] Porsev S G and Derevianko A 2003 *J. Chem. Phys.* **119** 844
- [67] Sahoo B K and Arora B 2013 *Phys. Rev. A* **87** 023402
- [68] Arora B and Sahoo B K 2014 *Phys. Rev. A* **89** 022511
- [69] Singh Y and Sahoo B K 2014 *Phys. Rev. A* **90** 022511
- [70] Mukherjee D and Pal S 1989 *Adv. Quant. Chem.* **20** 281
- [71] Sahoo B K 2010 *J. Phys. B* **43** 085005
- [72] Sahoo B K, Aoki T, Das B P and Sakemi Y 2016 *Phys. Rev. A* **93** 032520
- [73] Sahoo B K, Nandy D K, Das B P and Sakemi Y 2015 *Phys. Rev. A* **91** 042507
- [74] Sahoo B K and Das B P 2015 *Phys. Rev. A* **92** 052511
- [75] Sahoo B K 2015 *Phys. Rev. A* **92** 052506
- [76] Efelina V, Widiyanto E, Megasari K, Triyana K, Kusumaatmaja A, Rusydi A and Santoso I 2016 *AIP Conf. Proc.* **1755** 150008
- [77] Mukherjee B, Tseng F, Gunlycke D, Amara K K, Eda G and Simsek E 2015 *Opt. Mater. Express* **5** 447–55
- [78] Cai W and Shalae V M 2010 *Optical Metamaterials: Fundamentals and Applications* (New York: Springer)
- [79] Yan X, Zhu L, Zhou Y, Yiwen E, Wang L and Xu X 2015 *Appl. Opt.* **54** 6732–6
- [80] Miffre A, Jacquest M, Buchner M, Trenec G and Vigue J 2006 *Eur. Phys. J. D* **38** 353
- [81] Holmgren W F, Revelle M C, Lonij V P A and Cronin A D 2010 *Phys. Rev. A* **81** 053607
- [82] Brooks M D G N, Trubko R and Cronin A D 2016 *Atoms* **4** 3
- [83] Frindt R F and Yoffe A D 1963 *Proc. R. Soc. Lond. A* **273** 135269
- [84] Eda G, Yamaguchi H, Fujita T, Voiry D, Chen M and Chhowallam M 2011 *Nano Lett.* **11** 5111
- [85] Cheiwchanchamnangij T and Lambrecht W R 2012 *Phys. Rev. B* **85** 205302
- [86] Ramasubramaniam A 2012 *Phys. Rev. B* **86** 115409
- [87] Qiu D Y, da Jornada F H and Louie S G 2013 *Phys. Rev. Lett.* **111** 216805
- [88] Shen C C, Hsu Y T, Li L J and Liu H L 2013 *Applied Physics Express* **6** 125801